Luminescent Polyoxotungstoeuropate Anion-Pillared Layered Double Hydroxides


Keywords: Layered double hydroxides / Layered compounds / Polyoxometalates / Host–guest systems / Luminescence

Novel luminescent polyoxometalate anion-pillared layered double hydroxides (LDHs) were prepared by aqueous ion exchange of a Zn–Al LDH precursor in nitrate form with the europium-containing polyoxotungstate anions \([\text{EuW}10\text{O}36]^{9–}\), \([\text{Eu}(\text{BW}11\text{O}39)(\text{H}_2\text{O})_3]^{6–}\) and \([\text{Eu}(\text{PW}11\text{O}39)_2]^{11–}\). The host-guest interaction has a strong influence on the nature of the final intercalated species, as evidenced by elemental analysis, powder X-ray diffraction (XRD), infra-red (IR) and Raman spectroscopy, solid state magic-angle spinning (MAS) \(^{11}\text{B}\) and \(^{31}\text{P}\) NMR spectroscopy, and photoluminescence spectroscopy.

Synopsis, powder X-ray diffraction (XRD), infra-red (IR) and Raman spectroscopy, solid state magic-angle spinning (MAS) \(^{11}\text{B}\) and \(^{31}\text{P}\) NMR spectroscopy, and photoluminescence spectroscopy.

Introduction

Layered double hydroxides are an important class of ionic lamellar solids with the general formula \([\text{M}^{2+}_{1–x}\text{M}^{3+}_x(\text{OH})_2](\text{A}^m)^{x–/m}\cdot n\text{H}_2\text{O}(\text{M}^{2+} = \text{Mg}^{2+}, \text{Zn}^{2+}, \text{Ni}^{2+} \text{etc., M}^{3+} = \text{Al}^{3+}, \text{Cr}^{3+}, \text{Ga}^{3+} \text{etc})[1]\). The positively charged layers, containing divalent and trivalent cations in octahedral positions, are separated by charge balancing anions and water molecules. The water molecules are connected to both the metal hydroxide layers and the interlayer anions through extensive hydrogen bonding. A range of organic or inorganic guests may be incorporated into LDHs by either ion exchange, direct synthesis or hydrothermal reconstruction of calcined precursors[2,3]. In particular, intercalation chemistry has been explored with the aim of introducing catalytically active sites and photo- and electroactive species. Many different types of metal coordination compounds and oxometalates have been immobilized in LDHs, including phthalocyanines, cyanocomplexes, oxalate complexes and polyoxometalates (POMs)[4].

The first report of LDHs containing polyoxometalates concerned their use as exhaust gas and hydrocarbon conversion catalysts.[5] Since then, a variety of iso- and heteropolyanions with different nuclearities and structures (Keggin, Dawson, Preyssler, Finke) have been incorporated into the interlayer space of these materials.[6–18] Two factors assume considerable importance for the successful intercalation of polyoxometalates into an LDH compound. First, the heteropolyanions should carry sufficient charge in order to be spatially capable of compensating the host layer charge. Ideally, the charge should be higher than that necessary in order to increase the lateral separation of the anions in the interlayer and afford a material with microporosity. The second consideration is that the intercalated anion should be resistant to hydrolytic decomposition over the pH domain of stability of the LDH.

In recent years, heteropolyoxometalates incorporating lanthanide ions have excited interest because of their luminescence properties and potential use in lasers and luminescent materials.[19–23] As a result, the preparation of hybrid materials containing these species is now being investigated.[24–32] Francesconi and co-workers recently described Mg–Al LDHs intercalated by luminescent polyoxometalate anions with Dawson-like \([\text{Eu}(\text{H}_2\text{O})_3(\alpha_2\text{-P}_2\text{W}_{17}\text{O}_{61})]^{7–}\) and Keggin-like \([\text{Eu}_2(\text{H}_2\text{O})_4(\text{PW}_{11}\text{O}_{39})_2]^{6–}\) structures.[32] In the present work, new polyoxotungstoeuropate anion-pillared layered double hydroxides were prepared by anion exchange reactions of a Zn–Al LDH precursor in nitrate form. Reaction of the wet precursor with the species \([\text{EuW}_{10}\text{O}_{36}]^{9–}\), \([\text{Eu}(\text{BW}_{11}\text{O}_{39})(\text{H}_2\text{O})_3]^{6–}\) and \([\text{Eu}(\text{PW}_{11}\text{O}_{39})_2]^{11–}\) gave the final materials, which were characterized by powder X-ray diffraction, solid state NMR \((^{11}\text{B}, ^{27}\text{Al}, ^{31}\text{P})\), infrared and Raman spectroscopy. The luminescence properties of the pillared LDHs were investigated in detail and are compared with those of the precursor polyoxotungstoeuropate anions.

Results and Discussion

Synthesis of Polyoxotungstate Anion-Pillared Layered Double Hydroxides

The layered double hydroxide starting material, denoted as ZnAl[NO\(_3\)]\(_2\), was prepared using a standard coprecipi-
tion method with special attention paid to the avoidance of carbonate interference. Anion-exchange reactions were then carried out on an aqueous slurry of ZnAl[NO3]_x with the polyoxotungstoeuropate anions [EuW_{10}O_{36}]^{9–}, [Eu(BW_{11}O_{39})(H_2O)]^{6–} and [Eu(PW_{11}O_{39})_2]^{11–} (Table 1). Analysis of the final polyoxotungstoeuropate anion-pillared layered double hydroxides was performed by ICP-AES and the metal atom ratios are presented in Table 2. The Zn/Al ratio was about 1.7 (x = 0.37) for all of the materials and therefore unchanged compared with that for the ZnAl[NO3]_x precursor. Polyoxometalate (A_m) loadings were calculated for [Zn_{1−x}Al_x(OH)_2]_m(A_{m−}O)_n as a percentage of the theoretical maximum calculated from the ICP-determined mol fractions of aluminium.

Table 1. Representative reaction conditions and observed gallery heights for polyoxotungstoeuropate anion-pillared layered double hydroxides.

<table>
<thead>
<tr>
<th>Sample</th>
<th>n_m [mmol]</th>
<th>C_m [mol dm^{-3}]</th>
<th>T [°C]</th>
<th>G_{d003} [Å]</th>
</tr>
</thead>
<tbody>
<tr>
<td>ZnAl[EuW_{10}O_{36}]^9–</td>
<td>0.479</td>
<td>0.019</td>
<td>70</td>
<td>4.8</td>
</tr>
<tr>
<td>ZnAl[Eu(BW_{11}O_{39})(H_2O)]^{6–}</td>
<td>0.175</td>
<td>0.017</td>
<td>60</td>
<td>9.6</td>
</tr>
<tr>
<td>ZnAl[Eu(PW_{11}O_{39})_2]^{11–}</td>
<td>0.392</td>
<td>0.012</td>
<td>60</td>
<td>9.8</td>
</tr>
</tbody>
</table>

[a] Number of moles of polyoxometalate treated with precursor slurry containing 0.25 g clay material. [b] Polyoxometalate concentration. [c] Gallery height, calculated by subtracting the layer thickness (4.8 Å) from the d_{003} spacing.

Table 2. Metal atom ratios determined by ICP-AES for the polyoxotungstoeuropate anion-pillared layered double hydroxides.

<table>
<thead>
<tr>
<th>Sample</th>
<th>Zn</th>
<th>Al</th>
<th>W</th>
<th>Eu</th>
<th>L^{[a]}</th>
</tr>
</thead>
<tbody>
<tr>
<td>ZnAl[EuW_{10}O_{36}]^9–</td>
<td>1.73</td>
<td>1.00</td>
<td>1.02</td>
<td>0.09</td>
<td>92</td>
</tr>
<tr>
<td>ZnAl[Eu(BW_{11}O_{39})(H_2O)]^{6–}</td>
<td>1.68</td>
<td>1.00</td>
<td>1.44</td>
<td>0.13</td>
<td>79</td>
</tr>
<tr>
<td>ZnAl[Eu(PW_{11}O_{39})_2]^{11–}</td>
<td>1.77</td>
<td>1.00</td>
<td>1.26</td>
<td>0.05</td>
<td>63</td>
</tr>
</tbody>
</table>

[a] Polyoxometalate (A_m) loadings as a percentage of the theoretical maximum calculated for [Zn_{1−x}Al_x(OH)_2]_m(A_{m−}O)_n from the ICP-determined mol fractions of aluminium. The calculations assume that the only incorporated polyoxoanions are as indicated in the sample abbreviations.

Powder X-ray Diffraction

Figure 1 shows the powder XRD patterns obtained for the ZnAl[NO3]_x starting material and the pillared derivatives. Successful ion exchange of nitrate ions for POMs was revealed by the absence of reflections due to the starting material and the appearance of several new equally spaced (00l) harmonics with the (003) reflection having shifted to lower 2θ values. As is often observed for LDH-POM intercalates,[17] the first-order peaks for the pillared phases have very low intensity. This may be related to the high X-ray scattering power of the interlayer, out of phase with the brucite-like layer.[33] The positions of the (003) reflections are confirmed by the presence of the peaks at about 7.25 (006), 4.85 (009), 3.60 (0012) and 2.90 (0015) Å for the phases with d_{003} ≈ 14.5 Å [traces (c) and (d)] and 4.86 (006) and 3.26 (009) Å for the phase with d_{003} ≈ 9.6 Å [trace (b)]. The patterns also contain large and asymmetric (hkl) reflections in the 33–50° (2θ) range. This is characteristic of some disorder in the periodicity of the layers, such as a turbostratic effect or an intergrowth of the rhombohedral and hexagonal polytypes.[34] Table 1 gives the gallery heights, calculated by subtracting the layer thickness (4.8 Å) from the d_{003} spacing, for materials dried under reduced pressure in a vacuum desiccator. The anions [Eu(BW_{11}O_{39})(H_2O)]^{6–} and [Eu(PW_{11}O_{39})_2]^{11–} afforded pillared derivatives with a gallery height of about 10 Å. This layer separation is typical of LDHs pillared by Keggin-type or related structures and is consistent with a guest orientation in which the C_3 axis of the oxygen framework is perpendicular to the brucite-like layers.[8,11,12,18] The positions and relative intensities of the (00l) reflections in traces (c) and (d) are entirely in agreement with results reported in the literature for related compounds, for example a Zn_{2}Al-LDH pillared by [a-H_2W_{12}O_{40}]^{6–} anions.[8] A surprising result was obtained for the LDH-POM ZnAl[EuW_{10}O_{36}]^9–. The basal spacing of about 9.6 Å is only slightly larger than that measured for the precursor ZnAl[NO3]_x (8.94 Å). In the sodium tetrastromium salt NaSr_{4}[EuW_{10}O_{36}]·34.5H_2O the structure of the decatungstoeuropate anion consists of two W_5O_{18}^{6–} units (derived from the hexatungstate W_6O_{19}^{2–}).
by removal of one W=O group) linked by the europium ion.[35] The four former bridging oxygen atoms from each \( \text{W}_5\text{O}_{18}^{-6} \) unit form a square antiprism around Eu. The shortest dimension of the \([\text{EuW}_{10}\text{O}_{36}]^{9-}\) anion is about 10 Å (including van der Waals radii for the oxygen atoms). A basal spacing of at least 14.8 Å would therefore be expected for an LDH intercalate containing this anion. The observed basal spacing is much shorter and is in fact only compatible with the presence of two oxygen-atom layers in the interslab space. For example, a very similar powder XRD pattern was reported by Vaysse et al. for a NiCo LDH intercalated by \( \text{W}_2\text{O}_7^{-2} \) anions.[36] All of the diffraction patterns for the exchanged products contain a broad and asymmetric peak at \( 2\theta = 5-11° \). This is a common feature for LDH-POM composites and is often attributed to a quasi-crystalline \( \text{M}^{2+}/\text{M}^{3+} \) salt of the POM formed in low quantities,[11] although alternative origins have also been claimed.[37]

**Infrared and Raman Spectroscopy**

The infrared and Raman spectra of the polyoxotungstoeuropate anion-pillared layered double hydroxides showed that the exchange of nitrate anions for polyoxotungstate anions was almost complete (by the absence in the infrared spectra of the nitrate \( \nu_3 \) absorption band at 1384 cm\(^{-1}\), or in one case just a very weak IR band present) and also that there was no carbonate present in these materials (by the absence in the infrared spectra of the carbonate \( \nu_3 \) absorption band at 1370 cm\(^{-1}\)). The intercalation of polyoxoanions into the layered double hydroxide was clearly shown by the presence in the infrared and Raman spectra of several bands in the range 700–1200 cm\(^{-1}\), on top of the lattice vibrations of the host (Figure 2, Table 3), assigned to X–O stretch (X = B or P), the terminal W=O stretch (918–989 cm\(^{-1}\)), and the W–O–W corner or edge-shared stretching modes (700–900 cm\(^{-1}\).[38,39] For ZnAl[Eu(PW\(_{11}\text{O}_{39})_{2}^{-11-}] \) and ZnAl[Eu(BW\(_{11}\text{O}_{39})(\text{H}_2\text{O})_6^{-6} \) the similarity of the guest vibrations to those for the corresponding polyoxotungstoeuropate salts indicates that the structures of the core lacunary anions, i.e. \( \text{PW}_{11}\text{O}_{39}^{-7} \) and \( \text{BW}_{11}\text{O}_{39}^{-9} \), were unaffected by the ion-exchange reaction. In contrast, the bands

![Infrared spectra (transmission mode) for the LDH intercalates](image)

Table 3. Vibrational data for the polyoxotungstoeuropate anions and the respective anion-pillared layered double hydroxides.[a]

<table>
<thead>
<tr>
<th>Sample</th>
<th>Clay lattice and nitrate vibrations [cm(^{-1})]</th>
<th>Polyoxoanion stretching frequency assignment [cm(^{-1})]</th>
</tr>
</thead>
<tbody>
<tr>
<td>ZnAl[NO(_3)(^{-})]</td>
<td></td>
<td></td>
</tr>
<tr>
<td>Na(<em>2)[EuW(</em>{10}\text{O}_{36})]</td>
<td>1623 m, 1383 vs, 621 s, 560 s, 427 s</td>
<td></td>
</tr>
<tr>
<td>ZnAl[EuW(<em>{10}\text{O}</em>{36})](^{-9})</td>
<td>1624 m, 624 s, 561 s, 427 vs</td>
<td>932 vs, 968 vs, 946 vs, 918 s, 883 m, 784 s</td>
</tr>
<tr>
<td>K(<em>4)[Eu(BW(</em>{11}\text{O}_{39})(\text{H}_2\text{O})_3]</td>
<td>943 s, 963 sh, 941 vs</td>
<td>863 m, 830 vs, 790 s, 758 m</td>
</tr>
<tr>
<td>ZnAl[Eu(BW(<em>{11}\text{O}</em>{39})(\text{H}_2\text{O})_6] )</td>
<td>1623 m, 1384 w, 623 s, 560 m, 429 vs</td>
<td>909 m, 943 m, 969 vs, 900 sh, 952 vs, 987 vs, 973 s, 955 s, 909 m, 895 m, 898 sh, 987 vs, 973 s</td>
</tr>
<tr>
<td>K(<em>{11})[Eu(PW(</em>{11}\text{O}<em>{39})</em>{2}] )</td>
<td>1102 s, 1048 s, 1098 m, 1049 s, 1098 m, 1049 s, 1098 m, 1049 s, 1098 m, 1049 s</td>
<td>895 m, 898 sh, 867 m, 830 sh, 779 vs, 730 sh</td>
</tr>
</tbody>
</table>

[a] Infrared and Raman (in italics) data: vs, very strong; s, strong; m, medium; w, weak; sh, shoulder.
attributed to the guest species in ZnAl[Eu(W10O36)6] are significantly different from those exhibited by solid Na9[EuW10O36], for example the W=O stretch for the guest species is shifted by 14 cm⁻¹ and the W–O–W stretching modes present a set of maxima at wavenumbers that are different from those for Na9[EuW10O36]. These changes indicate that the intercalated species have a different structure, in agreement with the powder XRD data and the photoluminescence results discussed below.

### 11B, 27Al and 31P MAS NMR

Figure 3 shows the 11B MAS NMR spectrum of the pillared LDH ZnAl[Eu(BW11O39)(H2O)3]6. The spectrum consists of one main peak (with a broad low frequency shoulder) centred at δ = −6.8 ppm, with a full-width at half-maximum (fwhm) of about 580 Hz. For comparison, the potassium salt K5H[Eu(BW11O39)(H2O)3] displays a single sharp (fwhm 15 Hz) and symmetric resonance at δ = −7.4 ppm, while the spectrum of lacunary K6.5Na1.5[HBW11O39] (fwhm 130 Hz) consists of a second-order quadrupole powder pattern, showing that boron is in a distorted tetrahedral coordination. The corresponding lanthanopolyoxotungstoborate salt, K4[Eu(BW11O39)(H2O)]3, gives rise to a much broader peak (fwhm 440 Hz) at 0.1 ppm (Figure 3). As described in ref.31, the broadening of the resonance as compared with the monovacant Keggin polyoxotungstoborate is due to insertion of the europium ion into the polyoxoanion. Hence, the observation of the broad resonance at δ = −6.8 ppm in the 11B MAS NMR spectrum of the pillared LDH ZnAl[Eu(BW11O39)(H2O)3]6] suggests that the coordination of the europium ion to the polyoxotungstoborate was maintained after intercalation into the LDH host. Taking into account the results from powder XRD, elemental analysis (which gave the W/Eu molar ratio as 11:1), vibrational spectroscopy and photoluminescence studies (see below), we propose that the resonance at δ = −6.8 ppm is due to intercalated monomeric 1:1 [Eu(BW11O39)(H2O)]6 anions. The crystal structure of the potassium salt, K4[Eu(BW11O39)(H2O)], is unknown, but by comparison with the analogous Ce³⁺ compound it is likely that a polymeric structure exists in the solid-state, comprising Keggin-type anions bridged by Eu³⁺ cations. The difference in the chemical shifts for the pillared LDH (δ = −6.8 ppm) and the potassium salt (δ = 0.1 ppm) may therefore be attributed to the dissociation of the polymer into the monomeric building blocks, and the intercalation of these species into the host LDH. In support of this, it has previously been shown that polymeric lanthanopolyoxotungstosilicates dissociate in aqueous solution. As mentioned above, the peak at δ = −6.8 ppm in the 11B MAS NMR spectrum of the pillared LDH has a weaker low frequency shoulder. This may be due to a small amount of an insoluble K²/Zn³⁺/Al³⁺ salt of the POM [Eu(BW11O39)(H2O)]6– (related with the broad peak at 2θ = 5–11° in the powder XRD pattern).

The 31P MAS NMR spectrum of solid K11[Eu(PW11O39)2] presents a peak centred at 0.3 ppm with a high frequency shoulder at δ = −1.5 ppm (not shown). The presence of two peaks is probably due to slight asymmetry in the structure with respect to the two {PW11O39} moieties. An aqueous solution of this potassium salt exhibits one resonance at δ = −0.3 ppm, suggesting that the 1:2 structure is stable in solution and that the two phosphorus atoms are equivalent. A complicated 31P MAS NMR spectrum was obtained for the pillared LDH ZnAl[Eu(PW11O39)2]11, comprising several overlapping peaks between ca. 5 and −12 ppm (Figure 4). Spectral deconvolution gave four main components at −12.2, −10.1, −2.5 and 2.1 ppm in 1.0:1.0:5.3:1.9 intensity ratios. A much weaker peak was also found at δ = −7.7 ppm with a relative integration of 0.2. The major signal at δ = −2.5 ppm is assigned to intercalated [Eu(PW11O39)2]³⁻ anions. The two overlapping peaks at −12.2 and −10.1 ppm are attributed to uncoordinated [PW11O39]⁻² anions in slightly different environments, and the peak at ca. 2.1 ppm is assigned to co-intercalated 1:1 [Eu(PW11O39)(H2O)]³⁻ anions. If we exclude the weak resonance at δ = −7.7 ppm (which, at present, we are unable to assign) and assume that there are no other tungsten or europium-containing species present, the peak area integrated intensities allow a value of 22.3 to be estimated for the overall W/Eu molar ratio. This is in reasonable agreement with the value of 25 found by elemental analysis. All compounds were also characterized by 27Al MAS NMR spectroscopy. The spectra for the precursor ZnAl[NO3] and ZnAl[PO3] samples present a sharp reso-
Photoluminescence

Figure 5 shows the room-temperature photoluminescence excitation (PLE) spectra of the europium-containing polyoxotungstate materials monitored within the $^5D_0 \rightarrow ^7F_2$ lines. The spectra are mainly composed of a series of straight lines ascribed to intra-4f transitions. Only the spectrum of Na$_9$[EuW$_{10}$O$_{36}$] presents a large broad band in the UV spectral region with two main peaks around 270 and 320 nm. The origin of the PLE components may be related to ligand-to-europium charge-transfer (LMCT) transitions resulting from the interaction between the europium ion and the POM, in particular LMCT states associated with O/Eu and O/W transitions as already reported for other polyoxometalates containing Eu$^{3+}$ ions.[31, 45, 46] The higher intensity of the Eu$^{3+}$ straight lines indicate that the Eu$^{3+}$ ions are excited via direct excitation into the intra-4f$^6$ levels, and only for Na$_9$[EuW$_{10}$O$_{36}$] is the sensitized process comparable with direct excitation into the Eu$^{3+}$ levels.

The incorporation of the europium-containing polyoxotungstate into the Zn–Al LDH (to give the material designated as ZnAl[Eu(W$_{11}$O$_{39}$)$_2$]) induces the appearance of a low-intensity band centred around 292 nm. For the other two LDH compounds the intensity of this band is negligible. The higher intensity of the Eu$^{3+}$ lines for all of the ion-exchanged LDH materials indicates that the excitation occurs essentially via direct lanthanide excitation, rather than by an efficient sensitized process involving the lanthanide ion ligands.

The changes induced in the Eu$^{3+}$ local structures due to intercalation of the polyoxotungstate anions in the LDH host were further investigated by comparison of the respective emission (PL) spectra. The spectra shown in Figure 6 were obtained for different excitation wavelengths, selected according to the excitation spectra in Figure 5, namely direct excitation into the Eu$^{3+}$ intra-4f$^6$ levels ($^5D_0$, 395 nm), and within the UV band, at 320 nm for K$_{11}$[Eu(W$_{11}$O$_{39}$)$_2$]·nH$_2$O and 292 nm for the LDH materials. We will first consider the emission features of the polyoxotungstate salts. All of the spectra display the well-known Eu$^{3+}$ orange-red emission ascribed to transitions between the first excited state $^5D_0$ and the $^7F_{0-4}$ levels of the ground multiplet. No emission from the ligands could be detected. The profile of the PL lines, namely the energy, fwhm and number of components, depends on the sample, confirming that the Eu$^{3+}$ local environments in the three compounds are different. For K$_3$H[Eu(W$_{11}$O$_{39}$)(H$_2$O)$_3$] and K$_{11}$[Eu(PW$_{11}$O$_{39}$)$_2$]·nH$_2$O, the higher number of Stark components, in particular for the $^5D_0 \rightarrow ^7F_{1,2}$ transitions (3 and 4 lines, respectively), suggests that the metal ion local structure is characterized by a low symmetry group without an inversion centre according to the higher intensity of the $^5D_0 \rightarrow ^7F_2$ transition. The spectrum of Na$_9$[EuW$_{10}$O$_{36}$] has already been reported.[45] The compound prepared by us revealed a different spectrum with singular features, in particular the absence of the $^5D_0 \rightarrow ^7F_0$ transition and an abnormally high intensity for the $^5D_0 \rightarrow ^7F_4$ transition. A deeper investigation of these singular emission features lies...
beyond the scope of the present manuscript and will be discussed in a forthcoming study. For all of the polyoxotungstates, upon changing the excitation wavelength, no major variations are observed in the energy, fwhm and number of emission lines, suggesting that all of the metal ions occupy the same average local coordination site within each polyoxotungstate. This conclusion is further supported by the analyses of the decay curves monitored within the $^{5}D_{0} \rightarrow ^{7}F_{2}$ transition for an excitation wavelength around 395 nm. The decay curves are well reproduced by a single exponential function from which the $^{5}D_{0}$ lifetime was extracted (Table 4). For the $[\text{EuW}_{10}\text{O}_{36}]^{9-}$ anion the decay curve was also monitored for an excitation wavelength within the large broad band in the PLE spectrum, revealing a $^{5}D_{0}$ lifetime value similar to that obtained for direct excitation in the $^{5}L_{6}$ Eu$^{3+}$ level.

Figure 6. Room-temperature emission spectra of the polyoxotungstoeuropates (dotted lines) and the corresponding layered double hydroxide intercalates (solid lines): (a) K$_{4}$[Eu[BW$_{11}\text{O}_{39}$](H$_{2}$O)$_{3}$], ZnAl[Eu[BW$_{11}\text{O}_{39}$](H$_{2}$O)$_{6}$], (b) Na$_{9}$[EuW$_{10}\text{O}_{36}$]$_{6}$, ZnAl[EuW$_{10}\text{O}_{36}$], (c) K$_{1}[\text{Eu}(\text{PW}_{11}\text{O}_{39})_{2}]n\text{H}_{2}\text{O}$, ZnAl[Eu(PW$_{11}\text{O}_{39}$)$_{2}$]$^{11-}$ excited around A: 395 nm and B: 292–320 nm.

Table 4. $^{5}D_{0}$ lifetimes, $\tau$, monitored around 611 nm at different excitation wavelengths (292 and 395 nm) for the polyoxotungstoeuropate anions and the respective anion-pillared layered double hydroxides.

<table>
<thead>
<tr>
<th>Sample</th>
<th>$\tau$ [ms]</th>
<th>292 nm</th>
<th>395 nm</th>
</tr>
</thead>
<tbody>
<tr>
<td>ZnAl[Eu<a href="H$_%7B2%7D$O">BW$<em>{11}\text{O}</em>{39}$</a>$_{3}$]$^{6-}$</td>
<td>0.260 ± 0.002</td>
<td>0.255 ± 0.002</td>
<td></td>
</tr>
<tr>
<td>[Eu<a href="H$_%7B2%7D$O">BW$<em>{11}\text{O}</em>{39}$</a>$_{3}$]$^{6-}$</td>
<td>0.302 ± 0.001</td>
<td></td>
<td></td>
</tr>
<tr>
<td>ZnAl[EuW$<em>{10}\text{O}</em>{36}$]$^{5-}$</td>
<td>0.347 ± 0.003</td>
<td>0.289 ± 0.001</td>
<td></td>
</tr>
<tr>
<td>[EuW$<em>{10}\text{O}</em>{36}$]$^{5-}$</td>
<td>2.976 ± 0.002</td>
<td>2.929 ± 0.005</td>
<td></td>
</tr>
<tr>
<td>ZnAl[Eu[PW$<em>{11}\text{O}</em>{39}$]$_{2}$]$^{11-}$</td>
<td>0.324 ± 0.002</td>
<td>0.326 ± 0.003</td>
<td></td>
</tr>
<tr>
<td>[Eu[PW$<em>{11}\text{O}</em>{39}$]$_{2}$]$^{11-}$</td>
<td>and</td>
<td>and</td>
<td></td>
</tr>
<tr>
<td>ZnAl[Eu(PW$<em>{11}\text{O}</em>{39}$)$_{2}$]$^{11-}$</td>
<td>2.950 ± 0.118</td>
<td>3.050 ± 0.038</td>
<td></td>
</tr>
<tr>
<td>[Eu(PW$<em>{11}\text{O}</em>{39}$)$_{2}$]$^{11-}$</td>
<td>–</td>
<td>2.084 ± 0.002</td>
<td></td>
</tr>
</tbody>
</table>

The emission spectra of the LDH-POM intercalates obtained by direct excitation into the lanthanide ion levels resemble those of the precursor polyoxotungstoeuropate salts with the exception of ZnAl[EuW$_{10}\text{O}_{36}$]$^{9-}$, which presents an inversion of the relative intensities of the $^{5}D_{0} \rightarrow ^{7}F_{1,2}$ transitions, indicating that the Eu$^{3+}$ local coordination site in the ion-exchanged material is different from that in Na$_{9}$[EuW$_{10}\text{O}_{36}$], in particular the local symmetry group around the metal ion lost the inversion centre after incorporation in the LDH. This is consistent with the results from powder XRD and vibrational spectroscopy. For the remaining polyoxotungstates, minor changes were measured in the PL spectra. For ZnAl[Eu(PW$_{11}\text{O}_{39}$)$_{2}$]$^{11-}$, an increase was observed in the relative intensity of the $^{5}D_{0} \rightarrow ^{7}F_{2}$ transition, while for ZnAl[Eu(BW$_{11}\text{O}_{39}$)(H$_{2}$O)$_{6}$]$^{5-}$ an enlargement of the emission lines was detected. These minor changes in the PL spectra are compatible with changes in the Eu$^{3+}$ second coordination shell, induced by the LDH incorporation. However, upon decreasing the excitation wavelength to 292 nm, major changes in the PL profile were recorded, in particular for ZnAl[EuW$_{10}\text{O}_{36}$]$^{5-}$ and ZnAl[Eu(PW$_{11}\text{O}_{39}$)$_{2}$]$^{11-}$, which must be related with the presence of another Eu$^{3+}$ local site that was not observed in the precursor. This metal ion local coordination site must be related with the appearance of the large broad band in the PLE spectra. The spectrum of ZnAl[Eu(BW$_{11}\text{O}_{39}$)(H$_{2}$O)$_{6}$]$^{5-}$ with an excitation wavelength of 292 nm was omitted from Figure 6 (see part B) because it resembles that obtained at 395 nm, thus strongly suggesting a single Eu$^{3+}$ environment. At this point it should be noted that the PL results for this material do not preclude the existence of a minor second phase comprising a quasi-crystalline K$^{+}$Zn$_{2}$Al$^{11+}$ salt of the POM [Eu(BW$_{11}\text{O}_{39}$)(H$_{2}$O)$_{6}$]$^{5-}$, the existence of which is indicated by the powder XRD and $^{11}$B MAS NMR results.

In order to further investigate the number of distinct Eu$^{3+}$ local coordination sites in each material the PL and the lifetimes of the $^{5}D_{0}$ excited state were acquired at low-temperature for the same excitation wavelengths used to record the room-temperature emission. The number of Eu$^{3+}$ local coordination sites can be inferred through the analysis of the energy and fwhm of the $^{5}D_{0} \rightarrow ^{7}F_{0}$ transition, estimated by deconvoluting the spectrum assuming a Gaussian function. The analysis of the energy and fwhm of the $^{5}D_{0} \rightarrow ^{7}F_{0}$ transition is quite important, since this transition occurs between nondegenerated levels and its energy is usually correlated with the sum of the nephelauxetic effects arising from the Eu$^{3+}$ first neighbours.$^{[47]}$ The existence of more than one line or the variation of its energy and fwhm upon variation of the excitation wavelength suggest the presence of more than one Eu$^{3+}$ local coordination site. Figure 7 shows the low temperature PL spectra for the three LDH-POM samples.

For ZnAl[Eu(BW$_{11}\text{O}_{39}$)(H$_{2}$O)$_{6}$]$^{5-}$ the data collected at low temperature support the conclusion drawn from the room temperature PL spectra, i.e. that the europium ions occupy the same average local coordination site, since the spectra obtained under different excitation wavelengths are essentially the same. Moreover, the energy and fwhm of the $^{5}D_{0} \rightarrow ^{7}F_{0}$ transition is independent of the selected excitation wavelength (Table 5). The nature of this local environment must involve a low site symmetry group without

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**FULL PAPER**

**Luminescent Layered Double Hydroxides**

inversion centre according to the number of components for the $^5D_0 \rightarrow ^7F_{0,2}$ transitions (1, 3 and 4, respectively) and the higher intensity of the $^5D_0 \rightarrow ^7F_2$ lines, respectively.

For the other two LDH-POM samples, changes are observed in the number, energy, fwhm and relative intensity of the $^5D_0 \rightarrow ^7F_{0,2}$ components as the excitation wavelength increases from 292 to 395 nm. Increasing the excitation wavelength for ZnAl[EuW$_{10}$O$_{36}^{9-}$] induces a reduction of the fwhm and a blue-shift of the $^5D_0 \rightarrow ^7F_0$ transition, suggesting the presence of more than one type of first coordination sphere for the europium ions. Moreover, changes are also observed in the energy and relative intensity of components for the remaining transitions. The presence of at least two distinct local coordination sites for the Eu$^{3+}$ cations is also apparent for ZnAl[Eu(PW$_{11}$O$_{39}$)$_2^{11-}$]. Upon varying the excitation wavelength from 262 to 395 nm the fwhm and the energy of the $^5D_0 \rightarrow ^7F_0$ transition decrease. An increase in the relative intensity of the $^5D_0 \rightarrow ^7F_1$ transition and a decrease in the intensity of the $^5D_0 \rightarrow ^7F_2$ transition are also detected. These results support the interpretation of the $^{31}$P MAS NMR spectrum, that is, the co-intercalation of [Eu(PW$_{11}$O$_{39}$)$_2^{11-}$] and [Eu(PW$_{11}$O$_{39}$)(H$_2$O)$_6^{4+}$] anions.

The $^5D_0$ lifetime was measured at low temperature using two different excitation wavelengths (292 and 395 nm) and a monitoring wavelength around the more intense line of the $^5D_0 \rightarrow ^7F_2$ transition (not shown). Regardless of the selected excitation wavelength, the experimental data for ZnAl[Eu(BW$_{11}$O$_{39}$)(H$_2$O)$_6^{6-}$] are well reproduced by a single exponential function, revealing a lifetime value of 0.255±0.002 ms (Table 4), which further confirms the existence of a single Eu$^{3+}$ coordination site in this material. For ZnAl[EuW$_{10}$O$_{36}^{9-}$], the decay curve is also well reproduced by a single exponential, but different lifetime values of 0.347±0.003 and 0.289±0.001 were obtained for excitation wavelengths around 292 and 395 nm, respectively (Table 5).

The decay curves for ZnAl[Eu(PW$_{11}$O$_{39}$)$_2^{11-}$] revealed a two-exponential behaviour for the two selected excitation wavelengths, indicating lifetime values of 0.324±0.002 and 2.950±0.118 ms.

We can estimate the efficiency, $q$, of the $^3D_0$ Eu$^{3+}$ excited state. Assuming that only nonradiative and radiative processes are essentially involved in the depopulation of the $^3D_0$ state, $q$ can be defined as shown in Equation (1).

\[
q = \frac{k_r}{k_r + k_{nr}}
\] (1)

where $k_r$ and $k_{nr}$ are the radiative and the nonradiative transition probabilities, respectively.

The emission intensity, $I_i$, taken as the integrated intensity of the emission curves, for the $^3D_0 \rightarrow ^7F_{0,4}$ transitions, is expressed by Equation (2).

\[
I_{i \rightarrow j} = \int h\nu_{i \rightarrow j} A_{i \rightarrow j} N_i S_{i \rightarrow j}
\] (2)

where $i$ and $j$ represent the initial ($^3D_0$) and final ($^7F_{0,4}$) levels, respectively, $h\nu_{i \rightarrow j}$ is the transition energy, $A_{i \rightarrow j}$ corresponds to Einstein’s coefficient of spontaneous emission and $N_i$ is the population of the $^3D_0$ emitting level.\[The radiative contribution may be calculated from the relative intensities of the $^3D_0 \rightarrow ^7F_{0,4}$ transitions. The branching ratio for the $^3D_0 \rightarrow ^7F_{5,6}$ transitions must be neglected as they are not observed experimentally. Their influence on the depopulation of the $^3D_0$ excited state can therefore be ignored. Since the $^3D_0 \rightarrow ^7F_1$ transition can be considered as a reference, due to its dipolar magnetic nature, $k_r$ can be calculated as shown in Equation (3).

\[
k_r = A_{0 \rightarrow 4} \sum_{j=0}^{4} \frac{S_{0 \rightarrow j}}{S_{0 \rightarrow 1}} h\nu_{0 \rightarrow j}
\] (3)

Table 5. Energy, $E$, [cm$^{-1}$] and fwhm [cm$^{-1}$] of the $^3D_0 \rightarrow ^7F_0$ transition at different excitation wavelengths (292 and 395 nm) for the anion-pillared layered double hydroxides.

<table>
<thead>
<tr>
<th>Sample</th>
<th>292 nm</th>
<th></th>
<th>395 nm</th>
<th></th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>$E$</td>
<td>fwhm</td>
<td>$E$</td>
<td>fwhm</td>
</tr>
<tr>
<td>ZnAl[Eu(BW$<em>{11}$O$</em>{39}$)(H$_2$O)$_6^{6-}$]</td>
<td>17231.4±0.5</td>
<td>33±1</td>
<td>17231.1±0.1</td>
<td>34±1</td>
</tr>
<tr>
<td>ZnAl[EuW$<em>{10}$O$</em>{36}^{9-}$]</td>
<td>17242.2±0.4</td>
<td>42±1</td>
<td>17247.5±0.4</td>
<td>36±1</td>
</tr>
<tr>
<td>ZnAl[Eu(PW$<em>{11}$O$</em>{39}$)$_2^{11-}$]</td>
<td>17234.4</td>
<td>42±1</td>
<td>17228.4±0.2</td>
<td>36±1</td>
</tr>
</tbody>
</table>
whole spectrum; in vacuo \( A_{0.4} = 14.65 \, s^{-1} \).\(^{[48]} \) An effective refractive index of 1.5 was used. It was not possible to evaluate the \( q \) value for the LDH intercalates ZnAl-\([\text{EuW}_{10}\text{O}_{36}]^{9–}\) and ZnAl-[Eu\((\text{PW}_{11}\text{O}_{39})_{2}]^{1–}\), due to the presence of more than one local coordination site for the metal ions. Since it was not possible to selectively excite the emission ascribed to each Eu\(^{3+}\) ion, the emission intensity, \( I \), cannot be determined independently, thus preventing any \( 3D_0 \) quantum efficiency calculations.

The quantum efficiency value for the compound K\(_3\)H[Eu\((\text{BW}_{11}\text{O}_{39})(\text{H}_2\text{O})_3]\) is lower than those for the other two polyoxotungstoeuropate salts, essentially due to a higher nonradiative transition probability, that may be due to a higher number of coordinated water molecules (Table 6). Although the radiative transition probability for the LDH intercalate ZnAl[Eu\((\text{BW}_{11}\text{O}_{39})(\text{H}_2\text{O})_5]\) is higher, there is also an increase in the nonradiative transition probability, contributing to a slight decrease of the overall \( 3D_0 \) quantum efficiency.

### Conclusions

A new family of luminescent materials has been prepared by treatment of a Zn–Al layered double hydroxide in nitrate form with different polyoxotungstoeuropate anions. The photoluminescence studies, in conjunction with elemental analysis, powder XRD, IR and Raman spectroscopy, and solid state MAS NMR, allow us to construct a detailed picture of the possible nature of the intercalated anions. After ion exchange with an aqueous solution of K\(_3\)H[Eu\((\text{BW}_{11}\text{O}_{39})(\text{H}_2\text{O})_3]\), powder XRD, elemental analysis for W and Eu, vibrational and \(^{1}B \) MAS NMR spectroscopy strongly indicate that the guest species in the resultant material, ZnAl[Eu\((\text{BW}_{11}\text{O}_{39})(\text{H}_2\text{O})_5]\), are monomeric 1:1 [Eu\((\text{BW}_{11}\text{O}_{39})(\text{H}_2\text{O})_5]\) anions, which give rise to a gallery height of about 10 Å. The photoluminescence data acquired at room temperature and also at low temperature confirm the existence of a single Eu\(^{3+}\) coordination site in this material. Overall, the room-temperature excitation and emission spectra for the pillared LDH are not very different from those for the potassium salt, K\(_3\)H[Eu\((\text{BW}_{11}\text{O}_{39})(\text{H}_2\text{O})_3]\)]. Slight differences were, however, observed in the lifetimes and transition probabilities of the \( 3D_0 \) level. Ion exchange of the starting LDH compound with a solution of K\(_{11–}\) [Eu\((\text{PW}_{11}\text{O}_{39})_2]\) also gave a pillared derivative with a gallery height close to 10 Å. The photoluminescence data for this material show the presence of at least two distinct local coordination sites for the Eu\(^{3+}\) cations, which supports the interpretation of the \(^{31}P \) MAS NMR spectrum, that is, the assignment of two of the peaks to co-intercalated 2:1 [Eu\((\text{PW}_{11}\text{O}_{39})_2]\) and 1:1 [Eu\((\text{PW}_{11}\text{O}_{39})(\text{H}_2\text{O})_4]\) anions. The \(^{31}P \) MAS NMR spectrum also indicates the presence of uncoordinated [PW\(_{11}\text{O}_{39}]^{9–}\) anions. All three of these species could conceivably give rise to pillared LDHs with gallery heights around 10 Å. Deconvolution of the \(^{31}P \) MAS NMR spectrum showed that the 2:1 Eu-containing anion was the major species present. These results show that, regardless of the initial polyoxometalate species in solution, the nature of the final intercalated species can be different, presumably due to the strong host-guest interaction(s). This was especially evident for the anion [Eu\((\text{BW}_{11}\text{O}_{39})_3]\), which gave rise to a material containing intercalated anions with a quite different and, as yet, unknown structure. The observed basal spacing was only compatible with the presence of anions smaller than the deca-tungstometalate species, and the photoluminescence studies reveal the presence of more than one first coordination sphere for the lanthanide ions.

### Experimental Section

**Reagents and Materials:** Commercial materials were of reagent grade or better and used without further purification. Literature methods were used to prepare sodium or potassium salts of the polyoxotungstoeuropate anions [Eu\((\text{PW}_{11}\text{O}_{39})_3]\),\(^{[49]} \) [Eu\((\text{BW}_{11}\text{O}_{39})(\text{H}_2\text{O})_3]\),\(^{[31]} \) and [Eu\((\text{PW}_{11}\text{O}_{39})_2]\).\(^{[50]} \) The Zn–Al LDH starting material, ZnAl[NO\(_3\)], was prepared as described previously.\(^{[12]} \)

**Synthesis of Polyoxotungstoeuropate Anion-Pillared Layered Double Hydroxides:** Anion-exchange reactions with a series of europium-containing polyoxotungstate anions were carried out on an aqueous slurry of ZnAl[NO\(_3\)], using the method described by Evans et al. (Table 1).\(^{[12]} \) A typical reaction scale was 12.5 cm\(^3\) and involved dropwise addition of an aqueous solution of the polyoxometalate (up to 100% in excess over that theoretically required for complete exchange) to the ZnAl[NO\(_3\)] slurry. The mixtures were then stirred for 3 to 5 h in the temperature range of 60–70 °C (Table 1). Yields for approximately 0.25 g precursor clay: ZnAl[Eu\((\text{W}_{10}\text{O}_{36})_6]\), 0.36 g; ZnAl[Eu\((\text{BW}_{11}\text{O}_{39})(\text{H}_2\text{O})_5]\), 0.42 g; ZnAl[Eu\((\text{PW}_{11}\text{O}_{39})_2]\), 0.30 g.

**Instrumentation:** The Zn, Al, Eu, W and P contents were measured by Inductively Coupled Plasma-Atomic Emission Spectrometry (ICP-AES, Analytical Laboratories, University of Aveiro). FT-IR spectra were collected from KBr pellets (Aldrich, 99%+, FT-IR grade) with a Mattson 7000 FT-IR spectrometer. FT-Raman spectra were recorded with a Bruker RFS100/S FT-Raman spectrometer (Nd:YAG laser, 1064 nm excitation). \(^{11}B\), \(^{27}Al\) and \(^{31}P \) MAS NMR spectra were recorded with a Bruker Avance 400 spectrometer at, respectively, 128.37, 104.26 and 161.97 MHz, using spinning rates of 14 kHz. The \(^{31}P \) MAS NMR spectra were recorded with

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**Table 6. Quantum efficiency, \( q \) [%], experimental, \( k_{\exp} \) [ms\(^{–1}\)], radiative, \( k_{r} \) [ms\(^{–1}\)], and nonradiative, \( k_{nr} \) [ms\(^{–1}\)], transition probabilities of the \( 3D_0 \) level for the polyoxotungstoeuropate salts and the LDH intercalate ZnAl-[Eu\((\text{BW}_{11}\text{O}_{39})(\text{H}_2\text{O})_5]\).**

<table>
<thead>
<tr>
<th>Anion</th>
<th>( k_{\exp} )</th>
<th>( k_{r} )</th>
<th>( k_{nr} )</th>
<th>( q )</th>
</tr>
</thead>
<tbody>
<tr>
<td>([\text{PW}<em>{11}\text{O}</em>{39}]_3])</td>
<td>0.480</td>
<td>0.353</td>
<td>3.309</td>
<td>7.9</td>
</tr>
<tr>
<td>([\text{BW}<em>{11}\text{O}</em>{39})(\text{H}_2\text{O})_3])</td>
<td>0.153</td>
<td>0.319</td>
<td>2.069</td>
<td>7.6</td>
</tr>
<tr>
<td>([\text{BW}<em>{11}\text{O}</em>{39})(\text{H}_2\text{O})_5])</td>
<td>0.326</td>
<td>0.215</td>
<td>3.049</td>
<td>7.9</td>
</tr>
<tr>
<td>ZnAl-[Eu((\text{BW}<em>{11}\text{O}</em>{39})(\text{H}_2\text{O})_5])</td>
<td>32.0</td>
<td>39.2</td>
<td>7.9</td>
<td>7.9</td>
</tr>
</tbody>
</table>

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45° pulses and 50 s recycle delays. The $^{27}$Al MAS NMR spectra were recorded with 9° pulses and 1 s recycle delays. The $^{1}$B MAS NMR spectra were recorded with 15° pulses and 5 s recycle delays. Chemical shifts are quoted in ppm from BF$_4^-$ (C$_6$H$_5$)$_3$ (1$^1$B), Al(H$_2$O)$_6$($^{27}$Al) and 85% H$_3$PO$_4$ (31P). Powder XRD data were collected at room temperature with a Philips X’pert diffractometer operating at 40 kV and 50 mA using Cu-K$_a$ radiation. The diffractometer was equipped with a curved graphite monochromator, in a Bragg-Brentano para-focusing optics configuration. The ZnAl(NO$_3$)$_2$ sample was step-scanned in 0.05° 20 steps with a counting time of 1 s per step. For the other samples, a step size of 0.02° 20 was used with a counting time of 16 s per step. The room-temperature emission and excitation spectra of the LDHs were recorded using a Jobin Yvon-Spx spectrometer (HR, 460) coupled to a R928 Hamamatsu photomultiplier. A Xe arc lamp (150 mW) coupled to a Jobin Yvon monochromator (TRIX 180) was used as the excitation source. All of the spectra were corrected for the response of the detector.

The room-temperature photoluminescence of the POMs and the spectra and lifetimes recorded at low-temperature (10 K) were detected with a modular double grating excitation spectrofluorimeter with a TRIX 320 emission monochromator (Fluorolog-3, Jobin Yvon-Spex) coupled to a R928 Hamamatsu photomultiplier, in the front face acquisition mode. All of the photoluminescence spectra were corrected for optics and detection system response.

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