

Review article

Design of materials for solid oxide fuel cells, permselective membranes, and catalysts for biofuel transformation into syngas and hydrogen based on fundamental studies of their real structure, transport properties, and surface reactivity

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Abstract

Advances in design of materials for solid oxide fuel cells, oxygen and hydrogen separation membranes, and catalysts for biofuel conversion into syngas and hydrogen are reviewed. Application of new efficient techniques of material synthesis and characterization of their atomic-scale structure, transport properties, and reactivity allowed to develop new types of efficient cathodes and anodes for solid oxide fuel cells, asymmetric supported oxygen, and hydrogen separation membranes with high permeability and structured catalysts with nanocomposite active components demonstrating high performance and stability to coking in steam/autothermal reforming of biofuels.

Keywords: Nanocomposites, Synthesis, Solid oxide fuel cells, Supported membranes, Structured catalysts, Biofuels.

Introduction

Production of syngas and pure hydrogen is a key problem in the field of modern hydrogen energy. Design of efficient, inexpensive, and stable to coking catalysts for conversion of biogas and biofuels into syngas and hydrogen, as well as catalytic membrane reactors, is an urgent challenge in the sustainable and renewable energy field [1]. Another related problem is the design of solid oxide fuels cells (SOFCs) that could efficiently generate energy at decreased temperatures from hydrogen, syngas, and biofuels. Hence, mixed ionic-electronic conducting materials for SOFC electrodes and oxygen/hydrogen separation membranes with improved functional properties are to be designed as well.

In this review, recent advances in developing intermediate temperature SOFCs, oxygen and hydrogen separation membranes, and catalysts for the conversion of biofuels into syngas and hydrogen are reviewed. This includes analysis of methods for synthesis of mixed ionic-electronic conducting materials with required enhanced properties, atomic-scale factors controlling their performance and stability, as well as the progress in working characteristics of these devices.

Efficient methods for synthesis of materials

The solid-state reaction and solution combustion synthesis [1-8] are among the most widely used techniques. Intermediate mechanical activation for solid-state reaction helps to obtain highly active precursors allowing the use of short-time calcination [1-3]. A lot of oxide materials were obtained using solution combustion synthesis with different kinds of fuels and chelating agents [4-8]. Pechini technique allows obtaining dispersed powders with uniformity of the element distribution owing to a high level of mixing in a polymerized matrix [9]. Ultrasonic dispersion based on a cavitation effect allows producing fine powders with a low aggregation [1,10].

For dense ceramics, hot pressing [11,12], microwave, and radiation thermal sintering allow decreasing sintering temperatures and processing duration [13-16].

Nanosized catalysts can be obtained using coprecipitation, seeded growth, auto-redox, electrospinning, template methods, one-pot method, wetness impregnation, synthesis in supercritical media, electrophoretic synthesis, and so on. [17-24]. Chemically assisted electrodeposition was developed for highly active electrodes [25] (Figure 1). Hydrothermal [23,26,27] and spray drying [10,28,29] techniques are used as well. During the *in situ* exsolution, metal cations incorporated into the oxide form active metallic nanoparticles on its surface (Figure 2) in reducing atmospheres [30-32].

Novel materials for proton-conducting ceramic fuel cells

It was suggested to use oxides with multicomponent doping (high entropy oxides) for achieving high protonic conductivity. The first protonic conductivity studies of $\text{BaZr}_{0.2}\text{Sn}_{0.2}\text{Ti}_{0.2}\text{Hf}_{0.2}\text{Ce}_{0.2}\text{O}_{3-\delta}$, $\text{BaZr}_{0.2}\text{Sn}_{0.2}\text{Ti}_{0.2}\text{Hf}_{0.2}\text{Y}_{0.2}\text{O}_{3-\delta}$, $\text{BaZr}_{1/7}\text{Sn}_{1/7}\text{Ti}_{1/7}\text{Hf}_{1/7}\text{Ce}_{1/7}\text{Nb}_{1/7}\text{Y}_{1/7}\text{O}_{3-\delta}$, and $\text{BaZr}_{0.15}\text{Sn}_{0.15}\text{Ti}_{0.15}\text{Hf}_{0.15}\text{Ce}_{0.15}\text{Nb}_{0.15}\text{Y}_{0.10}\text{O}_{3-\delta}$ were accomplished [33]. The maximal protonic conductivity was achieved for $\text{BaZr}_{0.2}\text{Hf}_{0.2}\text{Sn}_{0.2}\text{Ti}_{0.2}\text{In}_{0.2}\text{O}_3$ (1.5×10^{-4} S/cm at 600 °C). Total conductivity of dense triple-doped solid solutions of barium cerate and zirconate $\text{BaCe}_{0.5}\text{Zr}_{0.2}\text{Y}_{0.1}\text{Yb}_{0.1}\text{Gd}_{0.1}\text{O}_{3-\delta}$ ceramics achieved 1×10^{-2} S/cm at 600 °C in wet air [34].

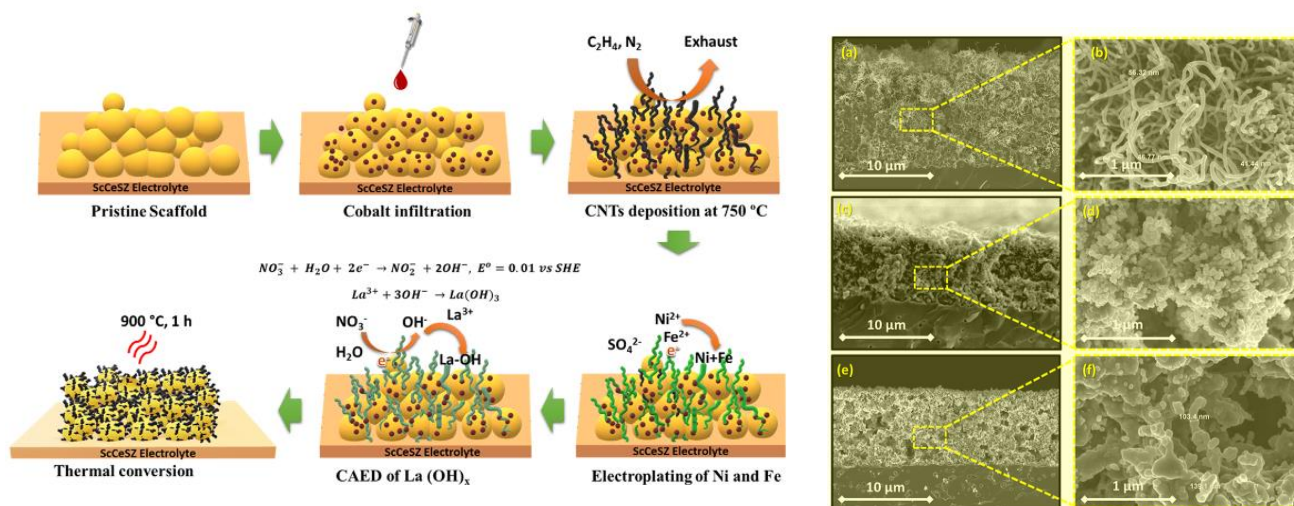


Figure 1. Preparation of nanofibrous composite cathode. Hybrid route including CAED for the preparation of a nanofibrous $\text{LaNi}_{0.6}\text{Fe}_{0.4}\text{O}_3\text{-Ce}_{0.9}\text{Gd}_{0.1}\text{O}_{1.95}$ (n-LNF-GDC) composite electrode (on the left) and evolution of the electrode microstructure during the process (on the right): (a) a carbon nanotube (CNT)-modified GDC scaffold; (b) a magnified image of the CNTs; (c) Ni-Fe-coated scaffold; (d) a magnified image of Ni-Fe coating; (e) a nanofibrous n-LNF-GDC cathode calcined at 900 °C; and (f) magnified image of the n-LNF-GDC cathode [19]. CAED, chemically assisted electrodeposition.



Figure 2. Demonstration of nickel exsolution effect [30].

High entropy BIMEVOXes are based on the parent $\text{Bi}_4\text{V}_2\text{O}_{11}$ with vanadium partially replaced by several metals [35]. Ionic conductivity of high entropy BIMEVOX is comparable to those for single-doped BIMEVOXes, they have an increased stability and the highest ionic transference numbers.

For calcium-doped layered perovskites $\text{BaNd}_{1-x}\text{Ca}_x\text{InO}_{4-x/2}$, total conductivity was 1-2 orders of magnitude higher than that for BaNdInO_4 [36].

Isotope exchange of oxygen

$^{18}\text{O}_2$ is commonly used for studying the oxygen mobility. However, oxygen substitution is often limited by oxygen dissociation on the surface, hence, lowering sensitivity to diffusion. A high surface exchange rate with C^{18}O_2 allows a comprehensive elucidation of the diffusion features. For C^{18}O_2 exchange with $\text{PrNi}_{0.5}\text{Co}_{0.5}\text{O}_{3-\delta}$, the response curve has two extrema (Figure 3) making evidence of at least two forms of oxygen in the bulk, whereas the fastest oxygen form is not discovered during the experiment with $^{18}\text{O}_2$. Such nonuniformity was found for most oxides, and composites and models were suggested to describe it [37-40].

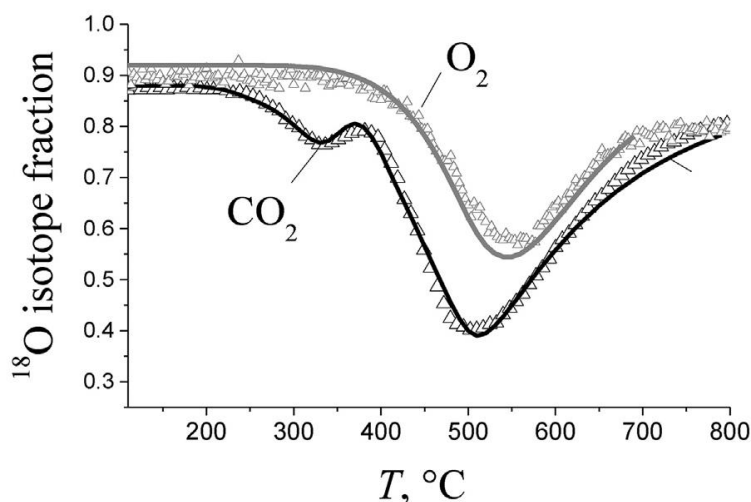


Figure 3. Temperature-programmed exchange with $^{18}\text{O}_2$ and C^{18}O_2 on $\text{PrNi}_{0.5}\text{Co}_{0.5}\text{O}_{3-\delta}$ in the flow reactor. Points — experiment, lines — calculation [37,38].

Symmetrical solid oxide fuel cells

Promising direction in SOFC technology is the development of symmetrical cells (SSOFCs) with two identical electrodes. The use of the same material for oxygen and fuel electrodes improves chemical and thermal compatibility of the cell components and contributes to simplification and cost-effectiveness of the fabrication process [41-44].

$\text{Sr}_2\text{Fe}_{1.5}\text{Mo}_{0.5}\text{O}_{6-\delta}$ (SFM) was proposed as a prospective symmetrical electrode material. The SFM was studied in oxidizing atmospheres to clarify kinetics of oxygen reduction and hydrogen oxidation in the anode conditions, as well as during redox cycling in an SSOFC [45-49]. Electrochemical activity of $\text{Sr}_{1.95}\text{Fe}_{1.4}\text{Ni}_{0.1}\text{Mo}_{0.5}\text{O}_{6-\delta}$ electrodes was higher than that of SFM in both oxidizing and reducing atmospheres, which can be explained by exsolution of nickel particles having high electrocatalytic activity [50,51].

The $\text{Pr}_2\text{NiO}_{4+\delta}$ catalyst effectively increases the rates of oxygen reduction and hydrogen oxidation. The rate of oxygen reduction after its introduction into the cathode increases owing to increase in the rate of oxygen exchange, whereas the enhanced rate of hydrogen oxidation is provided by the increased rate of its adsorbed molecule dissociation (Figure 4) [49].

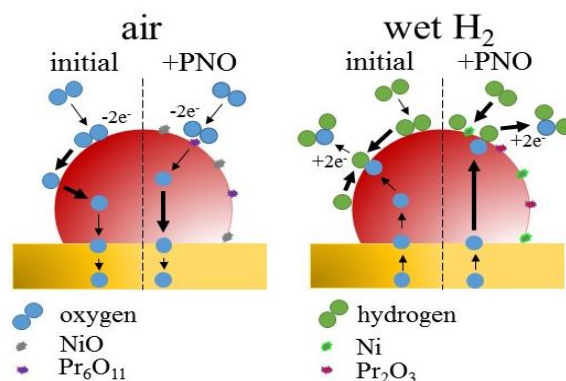


Figure 4. The pathways for oxygen reduction and hydrogen oxidation reactions before and after impregnation with the PNO catalyst [49]. PNO, Pr₂NiO_{4+δ}.

The application of this design to proton ceramic cells is only at the beginning of its development. New Nd_{1-x}Ba_xFe_{0.9}Mo_{0.1}O_{3-δ} materials (M = Cu, Ni; x = 0.4, 0.6) were investigated [52]. The reported results are among the first concerning the effective electrode operation in SSOFC's with proton-conducting electrolytes.

Reversible solid oxide cells

Reversible solid oxide cells are promising electrochemical devices which can function effectively both in the fuel cell and electrolysis mode [53,54]. The reversible solid oxide cells based on proton ceramic (rPCC) with a BaCe_{0.5}Zr_{0.3}Y_{0.1}Yb_{0.1}O_{3-δ} membrane generated a maximal power density and a hydrogen evolution rate. This complex approach allowed the main factors influencing the rPCC performance to be disclosed, and some approaches were proposed for future rPCCs' optimization. Because the ohmic resistance exceeds 50% of the total rPCCs' resistance in all studied modes, achievement of the next generation of cells operating below 600 °C with high performance and efficiency can be implemented through decreasing the electrolyte thickness to the 1-5 μm range [55-59].

Solid oxide fuel cell anodes

Metallic nickel in Ni-YSZ anodes serves as a catalyst for fuel oxidation and provides a path for electronic conduction. It is not suitable for hydrocarbon-fueled SOFCs because Ni catalyzes hydrocarbons cracking resulting in coking and also suffers from poisoning by sulfur and other contaminants. Significant efforts have been dedicated to the development of vanadium- and molybdenum-based components. Increasing Mo fraction in Ni-Mo catalysts significantly improves resistance to coking although reduces catalytic activity [60]. The Ni(11 wt.%)–Mo(3 wt.%)–YSZ reforming layer at the Ni-YSZ anode ensured a stable performance of the isooctane-fueled cell, whereas a similar cell without a reforming layer failed after 12 h [61]. The addition of molybdenum to the Ni–Ce_{0.5}Zr_{0.5}O₂ anode improved the electrical conductivity, nearly doubled power density, and ensured a stable operation [62]. The long-term (1737 h) stable performance of a cell with the Ni–Mo–CeO₂ anode fueled by H₂S and siloxane-contaminated biogas was demonstrated [63].

Perovskite-type AVO_3 and $AMoO_3$ ($A = Ca, Sr$) oxides attracted attention as prospective ceramic components owing to their high electronic conductivity under reducing conditions. Increasing molybdenum content in the B-sublattice of $CaV_{1-x}Mo_xO_{3-\delta}$ increases electrical conductivity and reduces thermal expansion [64]. Materials were stable in the pO_2 range corresponding to operation conditions but undergo oxidative decomposition at higher pO_2 . A partial substitution by transition metal with a more stable oxidation state can extend the stability domain. A compromise between a high electrical conductivity of $SrVO_3$ and thermodynamic stability of $SrTiO_3$ was demonstrated for $SrV_{1-y}Ti_yO_{3-\delta}$ [65]. Note the difference in the thermodynamic stability of V- and Mo-based perovskites under highly reducing conditions. Easier reducible molybdenum tends to exsolve at temperatures ≥ 1000 °C [64,66], whereas vanadium cations remain in a mixed 3+/4+ oxidation state even at 1500 °C [64].

$SrMo_{0.5}Ti_{0.5}O_{3-\delta}$ exhibits higher conductivity than that of the V-based counterpart [67]. This material showed good compatibility with electrolytes, good coking resistance, and sulfur tolerance [67]. The electrochemical activity of $SrMo_{0.5}Ti_{0.5}O_{3-\delta}$ can be improved by providing a minor cation deficiency in the B-sublattice [68]. The power density of cells with $Sr(Mo,Ti)O_{3-\delta}$ anodes fueled by H_2S -contaminated syngas can be increased ~2 times by the surface modification with Pd [67,68].

Oxidized $SrV_{0.5}Mo_{0.5}O_4$ can be easily reduced to highly conductive $SrV_{0.5}Mo_{0.5}O_{3-\delta}$ [69]. The reduction of the nickel-doped $SrV_{0.5}Mo_{0.5}Ni_{0.1}O_{4-\delta}$ precursor leads to exsolution of nickel covering the surface of perovskite: this results in the increase in electrical conductivity; a decrease in the electrode polarization resistance; a good electrocatalytic activity in H_2S -contaminated and hydrocarbon fuels; as well as a stable performance [69].

Asymmetric supported oxygen and hydrogen separation membranes

Conventional oxygen and hydrogen separation membranes consist of perovskites and noble metals or their alloys, respectively, lacking stability [1,38]. Using nanocomposites can help to avoid such drawbacks. Membranes based on $PrNi_{0.5}Co_{0.5}O_3$ - $Ce_{0.9}Y_{0.1}O_2$, $La_{0.3}Bi_{0.7}MnO_{3+\delta}$ - $Bi_{1.5}Y_{0.3}Sm_{0.2}O_2$, and $La_{5.5}WO_{11.25-\delta}$ - NiCu - demonstrated high oxygen/hydrogen permeation fluxes and stable performance in catalytic reactions [1,38,70,71].

Structured catalysts with nanocomposite-active components for biogas and biofuel conversion into syngas and hydrogen

The main problem for catalysts for biofuel reforming, especially for the methane dry reforming, is coking leading to deactivation, especially for those composed of inexpensive Ni on different supports [72-75]. The most efficient approaches to deal with this problem are based on the next concepts of design:

1. Suppress acidity of supports responsible for catalysis of reactions leading to coking.
2. Use oxide supports with a high oxygen mobility and reactivity containing transition and/or rare-earth cations reduced in the reaction conditions and able to activate carbon dioxide on the oxygen vacancies generating reactive oxygen species rapidly migrating to the metal-oxide interface and interacting with the activated fragments of fuels converting them into syngas.

3. Provide a strong metal-support interaction, which prevents metal sintering in the reaction conditions and ensures developed metal-support interface.
4. Use bimetallic catalysts, the best Ru-Ni alloy nanoparticles to prevent carbon nucleation.
5. Use supports with a core-shell structure (mesoporous MgAl_2O_4 with supported layers of perovskite, fluorite, or spinel oxides with high oxygen mobility and reactivity) to increase surface area, porosity, and thermal stability of catalysts.
6. Use heat-conducting substrates to support active components, thus providing efficient heat and mass transfer in the reactors.

Doping of alumina with Mg indeed effectively suppresses acidity (increases proton affinity) as demonstrated by the analysis of the shift of hydroxyls infra-red (IR) bands owing to CO adsorption [76]. This results in a higher stability of Ni-containing catalysts on these supports in the reactions of methane [72-77], ethanol [76], and other biofuels (glycerol, ethylacetate, and so on) [76,78] reforming owing to coking suppression.

For Ni catalysts on oxide supports with perovskite, fluorite, and spinel structures containing cations able to change their charge state in the reaction conditions (Mn, Cr, Fe, Pr, Ce, and so on), coking was suppressed owing to their high oxygen mobility and reactivity [75,76,79-83]. For such catalysts, the independent activation of reagents was reliably demonstrated by pulse experiments (including temporal analysis of products (TAP) pulse responses and pulse microcalorimetry) and SSITKA, steady state isotope transient kinetic analysis [75,80,84]. Estimation of oxygen self-diffusion coefficients for these catalysts demonstrated their high values providing a fast oxygen transport to the metal-support interface [75,80]. At the same time, for Ni/ $\text{La}_2\text{Zr}_2\text{O}_7$ catalyst without cations of variable charges in the support, SSITKA revealed that oxygen of the support is not involved in the methane dry reforming [85].

The most reliable method to characterize the metal/support interaction is Fourier-transformed infrared spectroscopy of the adsorbed CO [75,76,80]. Small sizes of metal particles owing to strong interaction with a support, their decoration by oxidic fragments, as well as the presence on the surface of metal alloy nanoparticles, atoms of each type mainly surrounded by atoms of another type results in domination in Fourier-transformed infrared spectroscopy spectra of adsorbed CO bands corresponding to terminal carbonyls (νCO 2000-2100 cm^{-1}), whereas bands corresponding to CO adsorbed on neighboring atoms of the same type (bridging carbonyls, $\nu\text{CO} < 2000 \text{ cm}^{-1}$) disappear. Domination of terminal carbonyls and a high intensity of their bands correlates with a high catalytic activity and coking stability of catalysts with Ni-Ru nanoparticles loaded on bulk fluorite, spinel and perovskite oxides [75,80], or Mg-doped alumina covered with their layers [76]. This is explained by prevention of carbon fragment nucleation on isolated Ni atoms and by suppression of carbon diffusion into the bulk of metal alloy particles, thus avoiding carbon nanotube nucleation at the metal-support interface [72,77].

Because catalysts based on bulk oxides with spinel, fluorite, and perovskite structure possessing a high oxygen mobility and reactivity have in general much lower specific surface area as compared with traditional supports ($\gamma\text{-Al}_2\text{O}_3$ and so on), a lot of work was devoted to design of core-shell structures by supporting reactive oxide layers on high surface area supports [75,76,78,84,86]. The most efficient combination was found to be mesoporous MgAl_2O_4 with supported layers of $\text{Ce}_{0.35}\text{Zr}_{0.35}\text{Pr}_{0.3}\text{O}_2$ fluorite promoted by Ru + Ni nanoparticles owing to optimized interaction between components, only moderately decreasing the reactive surface oxygen content and increasing its bonding strength [76,84].

Use of the ethylenediaminetetraacetic acid, EDTA-chelating agent in Ni supporting by impregnation on a ceria-loaded MgAl_2O_4 improved catalysts activity and stability in ethanol steam reforming owing to stronger metal-support interaction [78]. However, application of acetylacetone to load Ru + Ni on the mesoporous MgAl_2O_4 support precovered by $\text{Ce}_{0.35}\text{Zr}_{0.35}\text{Pr}_{0.3}\text{O}_2$ or to load $\text{PrNi}_{0.9}\text{Ru}_{0.1}\text{O}_3$ on a pure MgAl_2O_4 support resulted in excessively strong incorporation of Ni and Ru into support, hence, decreasing catalytic activity [76]. Hence, preparation procedures are to be optimized with a due regard for interaction of active components with the support at the impregnation step. Thus, for 50 wt.% $\text{Pr}_{0.15}\text{Sm}_{0.15}\text{Ce}_{0.35}\text{Zr}_{0.35}\text{O}_2$ + 50 wt.% $\text{LaMn}_{0.45}\text{Ni}_{0.45}\text{Ru}_{0.1}\text{O}_3$, nanocomposite formation of the perovskite phase from a polyester matrix with dispersed fluorite oxide provides the nanocomposite with a high specific surface area, developed interphases, and a high activity in EtOH (ethanol) reforming. In a one-pot synthesis method from a polyester containing all cations, the perovskite phase was not formed owing to La incorporation into the fluorite matrix, hence, a lower activity was demonstrated [81].

For any practical application, catalysts for conversion of biofuels into syngas are to be supported as thin layers on monolithic substrates, which allows minimization or even elimination of any heat and mass transfer limitations typical for granulated catalysts [75,76,87]. Supporting thin layers of active components based on a mesoporous MgAl_2O_4 support loaded with Ru + Ni/ $\text{Ce}_{0.35}\text{Zr}_{0.35}\text{Pr}_{0.3}\text{O}_2$ on a FeCr alloy honeycomb substrate allowed to provide a stable performance and high yields of syngas in dry reforming of natural gas, steam reforming of ethanol, and oxy-steam reforming of ethylacetate at contact times several times lower than for catalysts on the same substrate loaded with catalysts based on bulk-doped Ce-Zr-O fluorite or traditional alumina supports [75,76].

Conclusions and outlook

New methods of materials synthesis for SOFC, permselective membranes, and catalysts for biofuel conversion apparently open the ways to improve their functional properties and achieve required characteristics of devices constructed on their bases. Though additional research is clearly required, especially in the case of SOFCs with proton-conducting electrolytes, the progress made in the last two years is apparent. From the fundamental point of view, the most important are results related to design of mixed ionic-electronic conductors via multiple doping or preparation of nanocomposites with a core-shell structure. Among the characterization methods, a great progress was achieved in the oxygen isotope exchange methods for estimation of oxygen self-diffusion coefficients and SSITKA to elucidate kinetic features of catalytic reactions.

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